

# **PERFORMANCE ANALYSIS OF SINGLE STEP CO-FIRED SOLID OXIDE FUEL CELLS (SOFCs)**

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# OUTLINE

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- ❖ **Introduction**
- ❖ **Single Step Co-firing Process Development**
- ❖ **Polarization Modeling**
- ❖ **Polarization Analysis**
- ❖ **Fuel Utilization Test**
- ❖ **Advanced Cathode Investigation**
- ❖ **Summary & Future Work**

# INTRODUCTION

## ❖ Major Challenges for Commercialization of SOFCs

**High Manufacturing Costs**  
(SOFC System Costs per Unit of Power (\$/kW))



**Single Step Co-Firing Process**  
with Conventional Material System

**Reduced Cell Performance at Low Operating Temperature** (Interconnects, Sealing, etc.)



**Polarization Modeling: Systematic Analysis of Cell Performance and Polarization Losses**

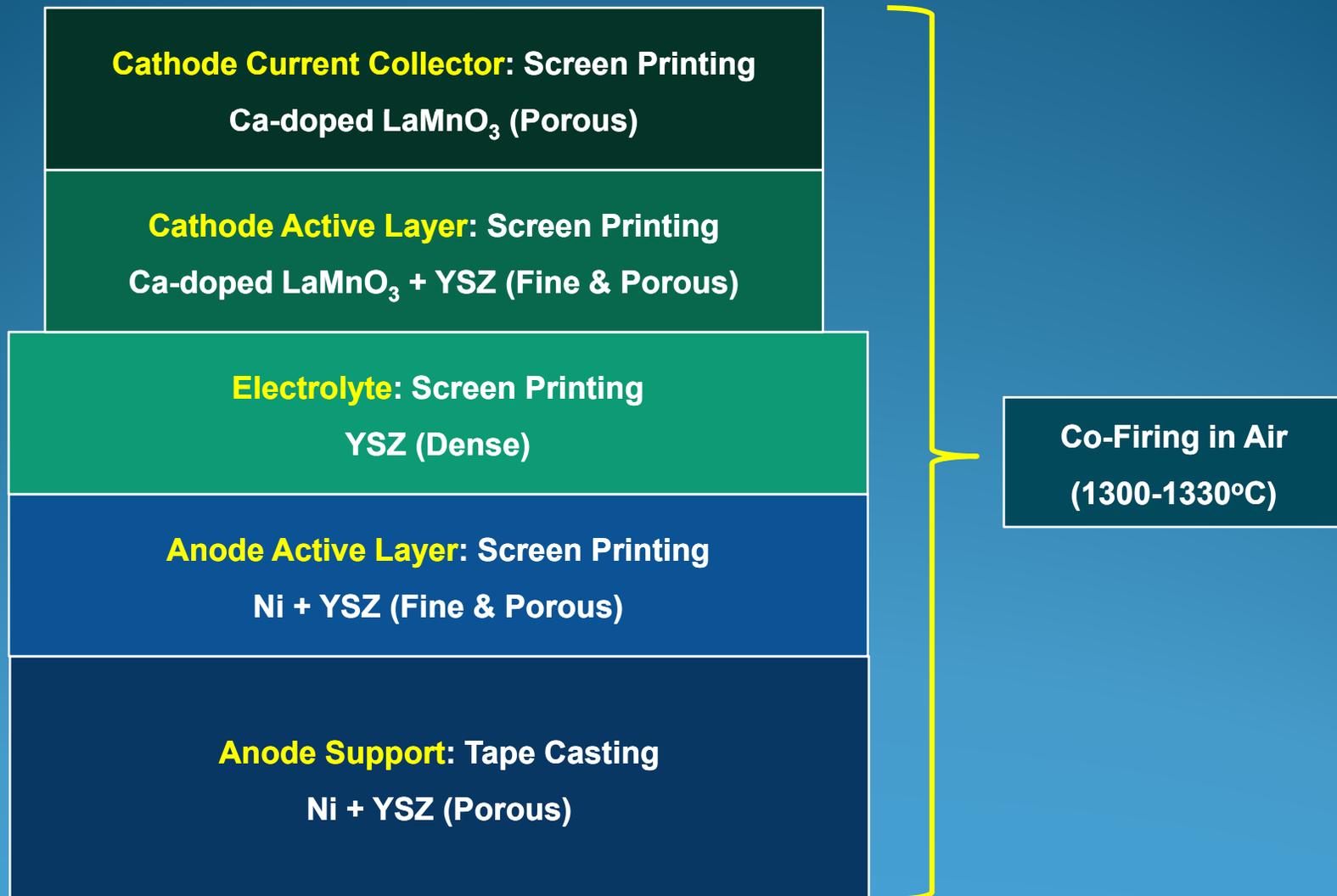
## ❖ Research Goals

- Manufacture SOFCs with the Lowest Manufacturing Costs  
by **Single Step Co-Firing of the Entire SOFCs**
- Achieve the Highest Cell Performance and Lower Operating Temperature  
by **Optimization of Materials, Microstructures, and Process Parameters using Polarization Modeling**



**Minimize \$/kW**

# SINGLE STEP CO-FIRING PROCESS



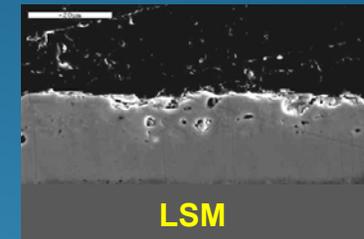
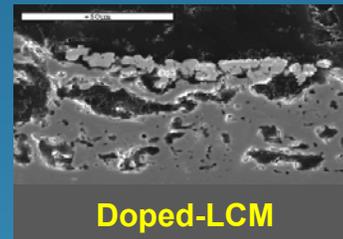
# PROCESS DEVELOPMENT

- ❖ Lowered Electrolyte Sintering Temperature
  - Density of YSZ : **Without Sintering Aid: ~ 94%, With Sintering Aid: ~ 99+% @1300°C**

- ❖ Matched Thermal Expansion Coefficients and Sintering Shrinkages

- ❖ Developed Refractory Cathode Composition

- **Doped-(La,Ca)MnO<sub>3</sub>**

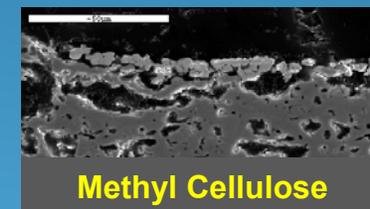


- ❖ Optimized **Thicknesses** and **Porosities** of Electrodes

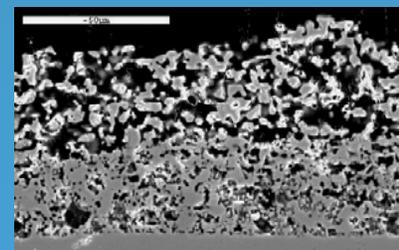
- ❖ Optimized **Particle Sizes** of Initial Powders

- ❖ Evaluated Pore Former Material

- **Carbon Black**

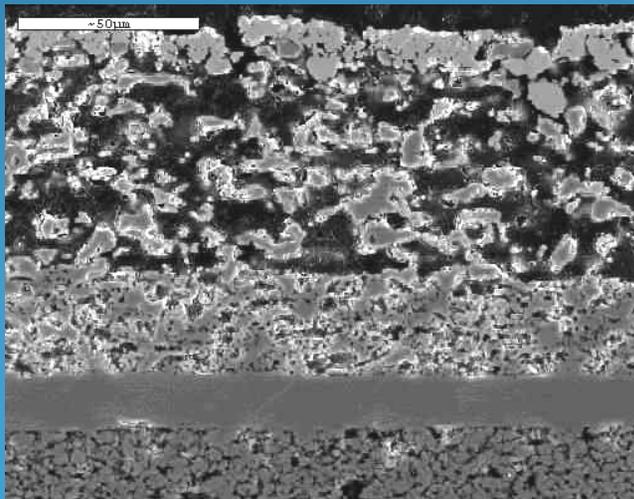
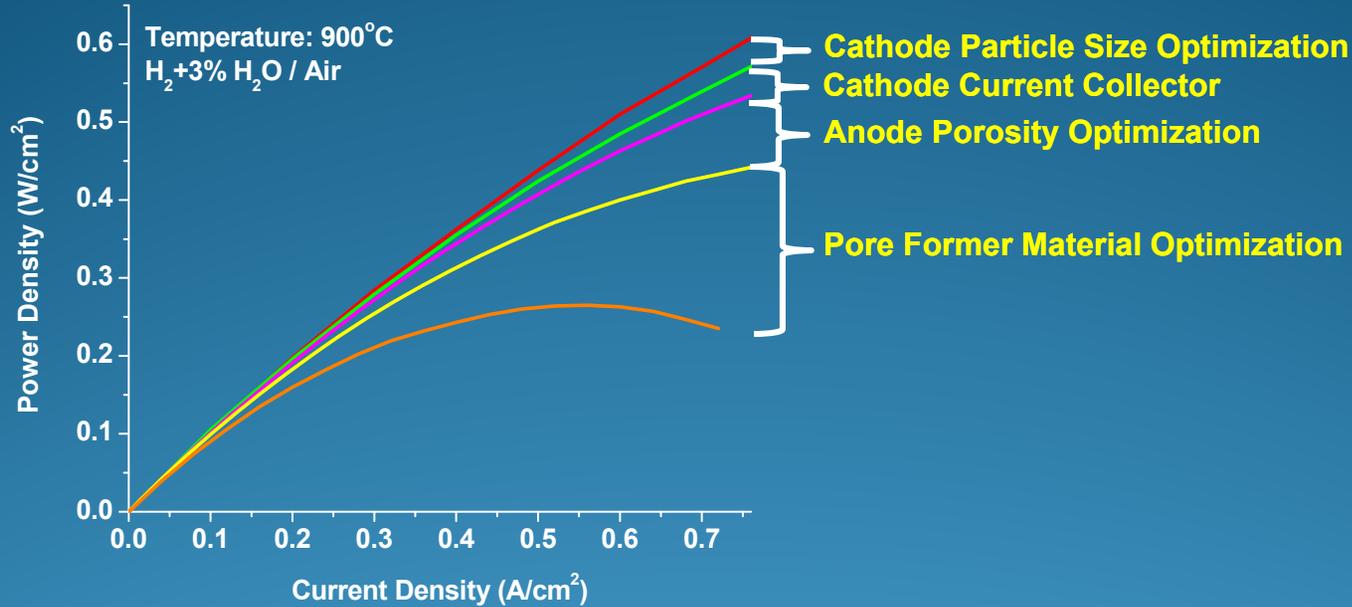


- ❖ Employed **Cathode Current Collector**



**Cathode Current Collector**  
: (La,Ca)MnO<sub>3</sub>  
50µm thick, 50% porous

# PROCESS DEVELOPMENT



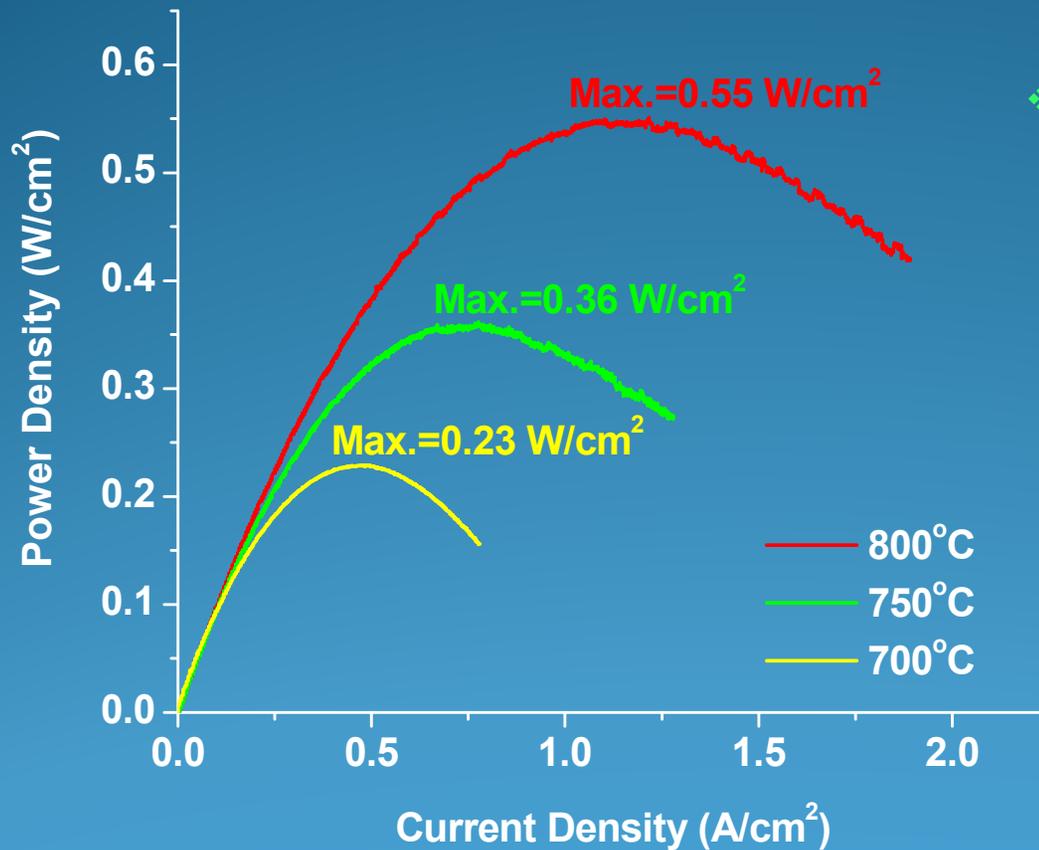
**Cathode Current Collector:**  
LCM, 50µm thick, 50% porous

**Cathode Active Layer:**  
LCM-YSZ, 30µm thick, 31% porous

**Electrolyte:** YSZ, 15µm thick

**Anode Support:**  
Ni-YSZ, 850µm thick, 32% porous

# BASELINE CELL: POWER DENSITY



❖ **Test Condition**

- Temperature: 700~800°C
- Fuel: 97% H<sub>2</sub>+3% H<sub>2</sub>O
- Oxidant: Air

# POLARIZATION MODEL: THEORY

$$E_C = E_0 - iR_i - \eta_{act} - \eta_{conc}$$

■ **Activation Polarization ( $\eta_{act}$ )\*** :  $i = i_0 \exp\left(\frac{\alpha n_e \eta_{act} F}{RT}\right) - i_0 \exp\left(-\frac{(1-\alpha) n_e \eta_{act} F}{RT}\right)$  ( $\alpha=1/2, n_e=2$ )

$$\eta_{act} = \frac{RT}{F} \ln\left\{\frac{1}{2}\left[\left(\frac{i}{i_0}\right) + \sqrt{\left(\frac{i}{i_0}\right)^2 + 4}\right]\right\} \quad (\text{Anode + Cathode})$$

■ **Concentration Polarization ( $\eta_{conc}$ )\*\*** :  $\eta_{conc} = -\frac{RT}{4F} \ln\left(1 - \frac{i}{i_{cs}}\right) - \frac{RT}{2F} \ln\left(1 - \frac{i}{i_{as}}\right) + \frac{RT}{2F} \ln\left(1 + \frac{p_{H_2}^o i}{p_{H_2O}^o i_{as}}\right)$

$$E_C = E_0 - iR_i - \frac{RT}{F} \ln\left\{\frac{1}{2}\left[\left(\frac{i}{i_0}\right) + \sqrt{\left(\frac{i}{i_0}\right)^2 + 4}\right]\right\} + \frac{RT}{4F} \ln\left(1 - \frac{i}{i_{cs}}\right) + \frac{RT}{2F} \ln\left(1 - \frac{i}{i_{as}}\right) - \frac{RT}{2F} \ln\left(1 + \frac{p_{H_2}^o i}{p_{H_2O}^o i_{as}}\right)$$

Open Circuit  
Potential

Ohmic  
Polarization

Activation  
Polarization

Concentration  
Polarization (Cathode)

Concentration  
Polarization (Anode)

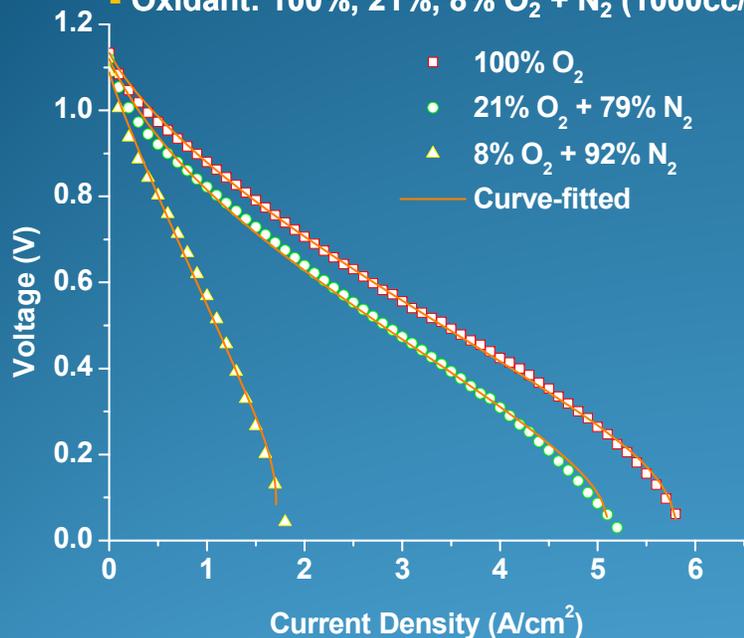
\* P.W.Li, M.K.Chyu, *J. Heat Transfer*, vol.127, 1344 (2005)

\*\* J.W.Kim, A.V.Virkar, K-Z.Fung, K.Mehta, S.C.Singhal, *J.Electrochem.Soc.*146 (1) (1999) 69

# POLARIZATION MODELING: CURVE-FITTING (EXAMPLE)

## Test Condition

- Temperature: 800°C
- Fuel: 97% H<sub>2</sub>+3% H<sub>2</sub>O (300cc/min): Fixed
- Oxidant: 100%, 21%, 8% O<sub>2</sub> + N<sub>2</sub> (1000cc/min)



Fitting Parameters	100% O <sub>2</sub>	21% O <sub>2</sub> + 79% N <sub>2</sub>	8% O <sub>2</sub> + 92% N <sub>2</sub>
$R_i$ ( $\Omega \cdot \text{cm}^2$ )	0.082	0.082	0.082
$i_o$ (A/cm <sup>2</sup> )	0.52	0.28	0.097
$i_{as}$ (A/cm <sup>2</sup> )	5.77	5.77	5.77
$i_{cs}$ (A/cm <sup>2</sup> )	-	5.42	1.77

Assumption:  $i_{cs} \gg i_{as}$  and  $\eta_{conc,c} \ll \eta_{conc,a}, \eta_{act}$  with 100% O<sub>2</sub>

Anode Limiting Current  $i_{as} = \frac{2 F p_{H_2}^o D_{H_2-H_2O}^{eff}}{R T l_a}$

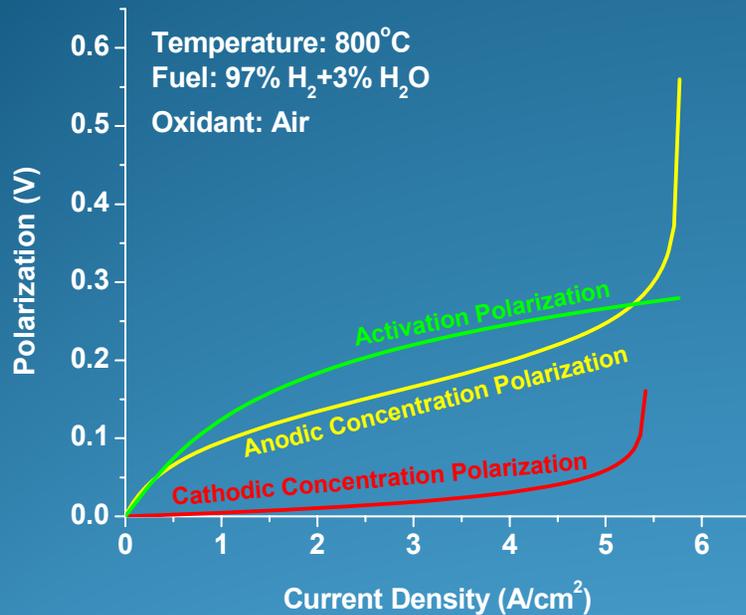
$D_{H_2-H_2O}^{eff} = 0.23 \text{ cm}^2/\text{s}$

Cathode Limiting Current  $i_{cs} = \frac{4 F p_{O_2}^o D_{O_2-N_2}^{eff}}{(\frac{p - p_{O_2}^o}{p}) R T l_c}$

$D_{O_2-N_2}^{eff} = 0.037 \text{ cm}^2/\text{s}$

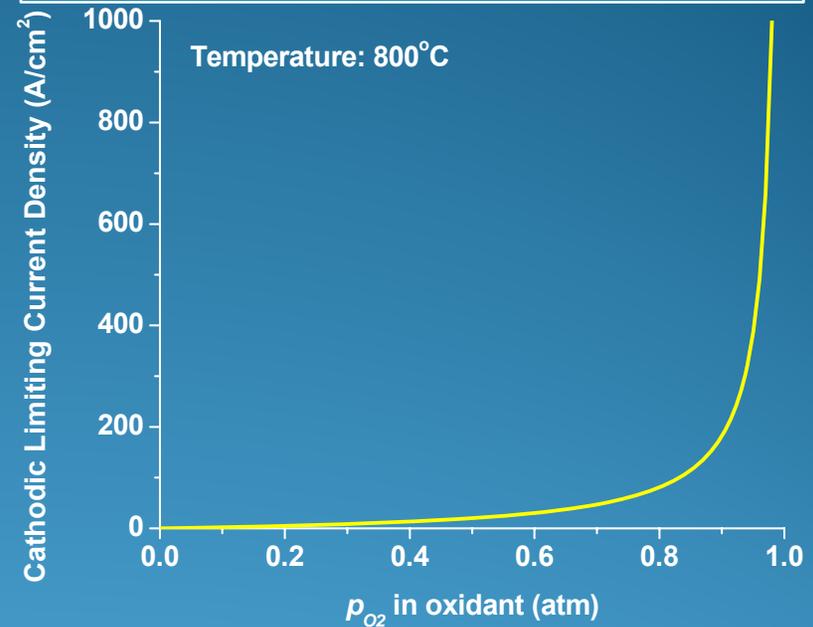
# POLARIZATION MODELING: VERIFICATION OF ASSUMPTION

## Various Polarization Losses @800°C with Humidified Hydrogen (3% H<sub>2</sub>O) and Air



▪  $\eta_{conc,c} \ll \eta_{conc,a}$  and  $\eta_{act}$  except at high current density near  $i_{cs}$

## Cathodic Limiting Current Density @800°C as a Function of $p_{O_2}$ in Oxidant



▪  $i_{cs}$  rapidly increases as  $p_{O_2}$  increases and approaches 100% O<sub>2</sub>

❖ Consistent with assumption that  $\eta_{conc,c}$  is negligible compared to other polarization losses when oxygen at the cathode nears 100%

# POLARIZATION MODELING & IMPEDANCE SPECTROSCOPY

- Activation Polarization Resistance at OCV

$$R_{act} = \frac{d\eta_{act}}{di} = \frac{RT}{F} \cdot \frac{1}{\sqrt{i^2 + 4i_0^2}}$$

$$R_{act}|_{i \rightarrow 0} = \frac{RT}{2Fi_0}$$

- Anodic Concentration Polarization Resistance at OCV

$$R_{conc,a} = \frac{d\eta_{conc,a}}{di} = \frac{RT}{2F} \left[ \frac{1}{i_{as} \left(1 - \frac{i}{i_{as}}\right)} + \frac{p_{H_2}^o}{p_{H_2O}^o} \cdot \frac{1}{i_{as} \left(1 + \frac{p_{H_2}^o}{p_{H_2O}^o} \cdot \frac{i}{i_{as}}\right)} \right]$$

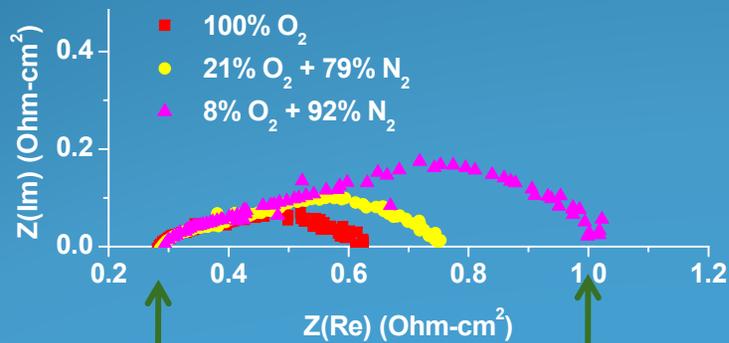
$$R_{conc,a}|_{i \rightarrow 0} = \frac{RT}{2F} \cdot \frac{1}{i_{as}} \left( 1 + \frac{p_{H_2}^o}{p_{H_2O}^o} \right)$$

- Cathodic Concentration Polarization Resistance at OCV

$$R_{conc,c} = \frac{d\eta_{conc,c}}{di} = \frac{RT}{4F} \cdot \frac{1}{i_{cs} \left(1 - \frac{i}{i_{cs}}\right)}$$

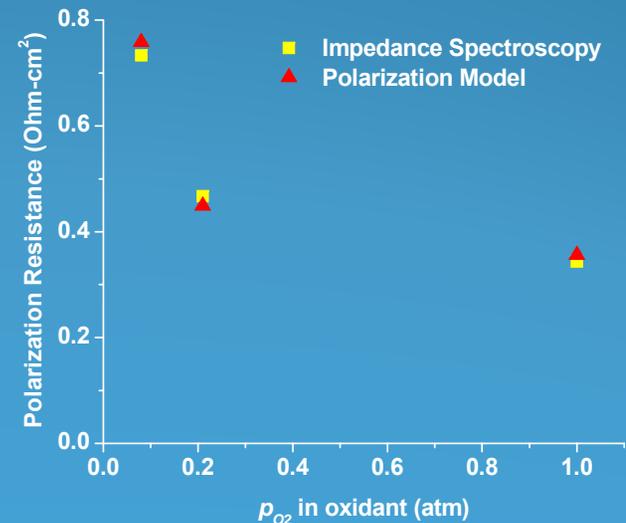
$$R_{conc,c}|_{i \rightarrow 0} = \frac{RT}{4F} \cdot \frac{1}{i_{cs}}$$

## Impedance Spectroscopy @800°C

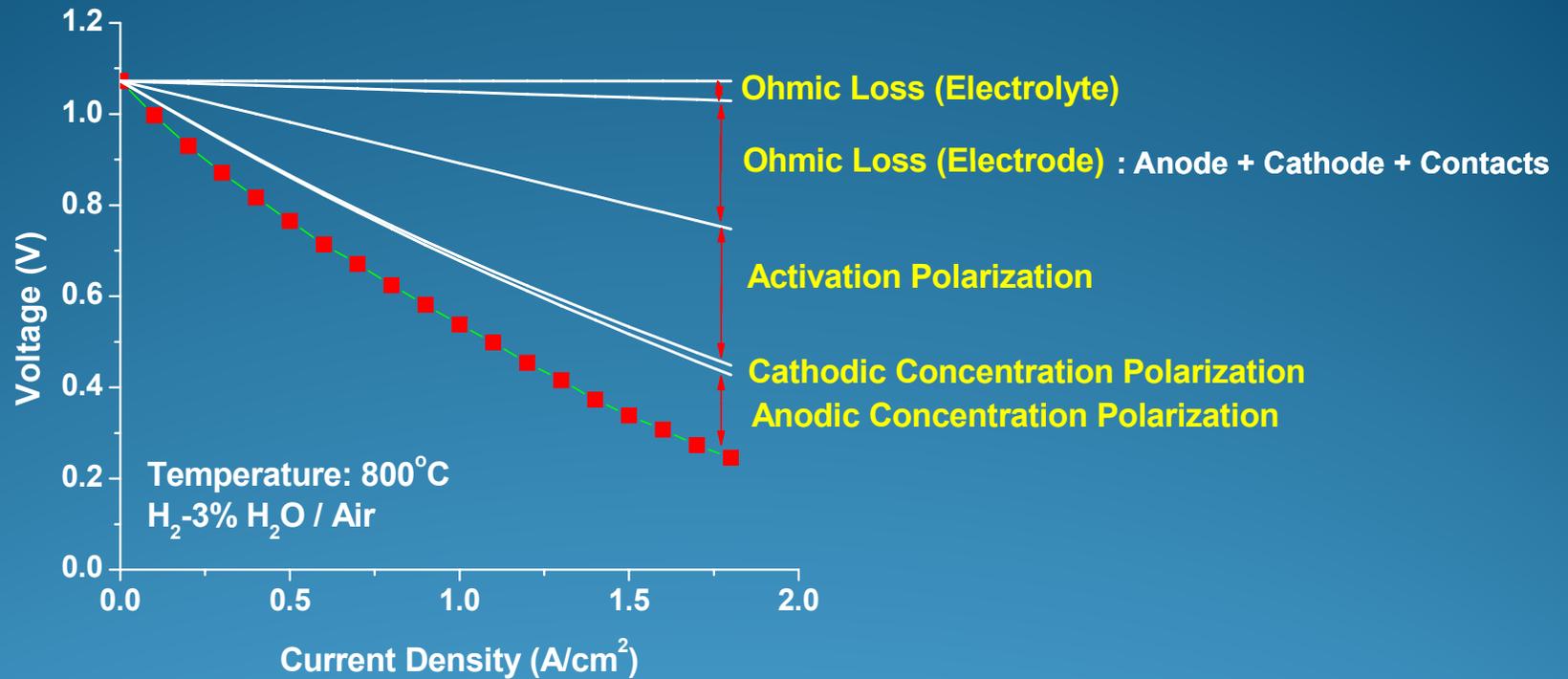


High Frequency Intercept =  $R_{\Omega}$

Low Frequency Intercept =  $R_{polar} + R_{\Omega}$



# POLARIZATION ANALYSIS #1: BASELINE CELL

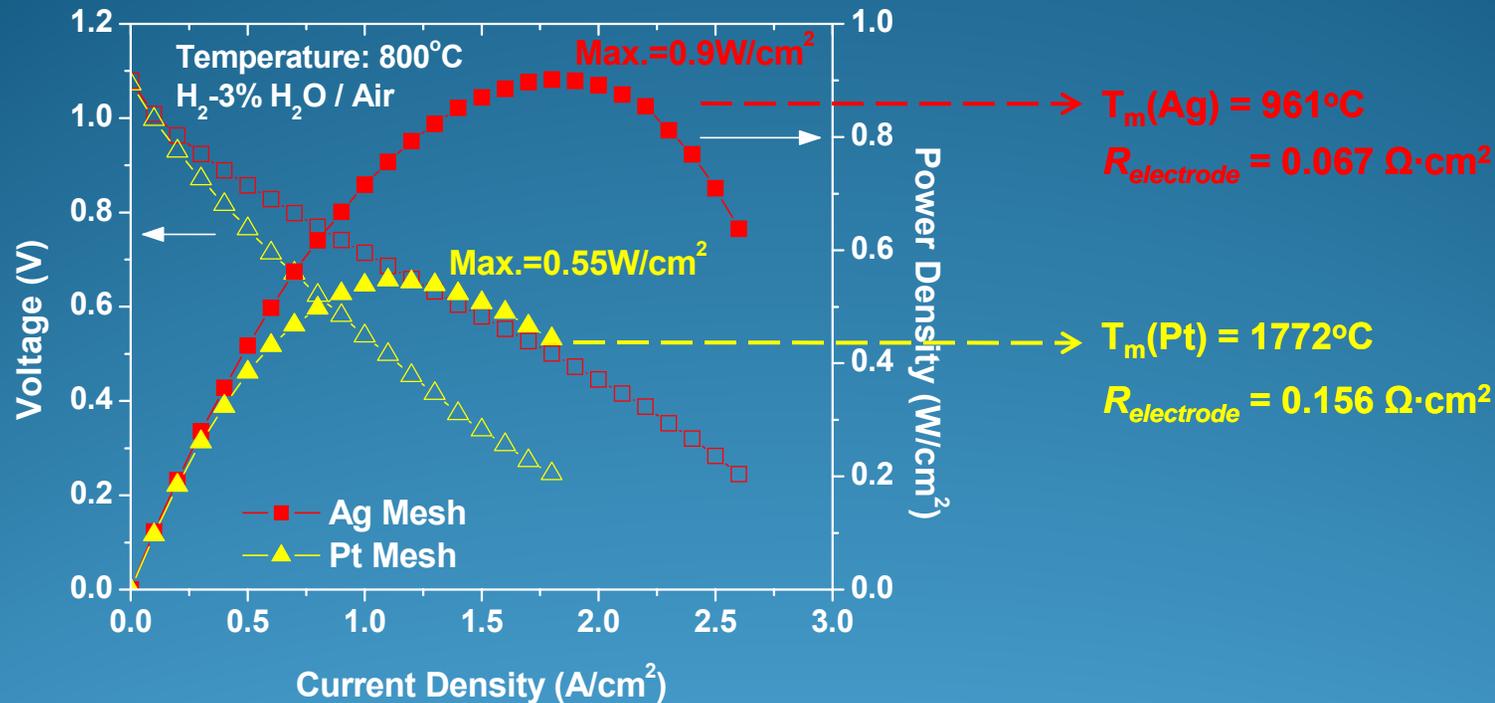


## ❖ Dominant Polarization Losses

- Ohmic Loss (Electrode)
- Activation Polarization (Cathode)
- Concentration Polarization (Anode)

# POLARIZATION ANALYSIS #2: CONTACT RESISTANCE

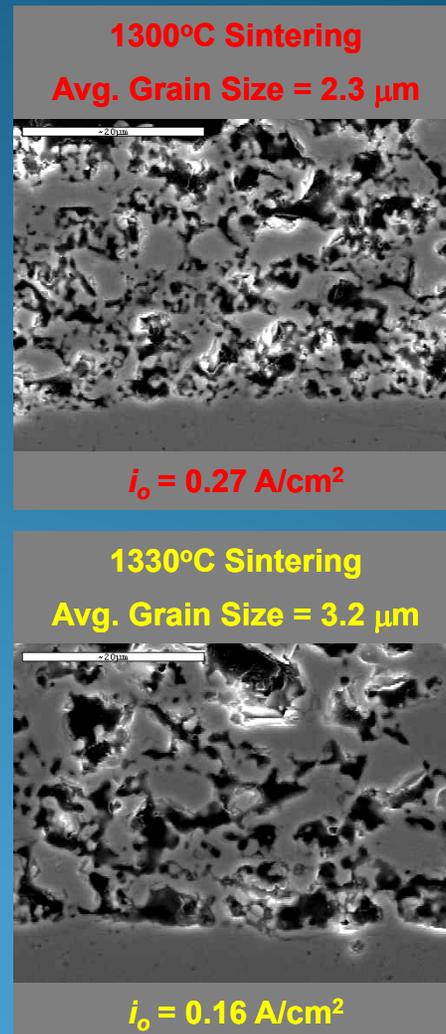
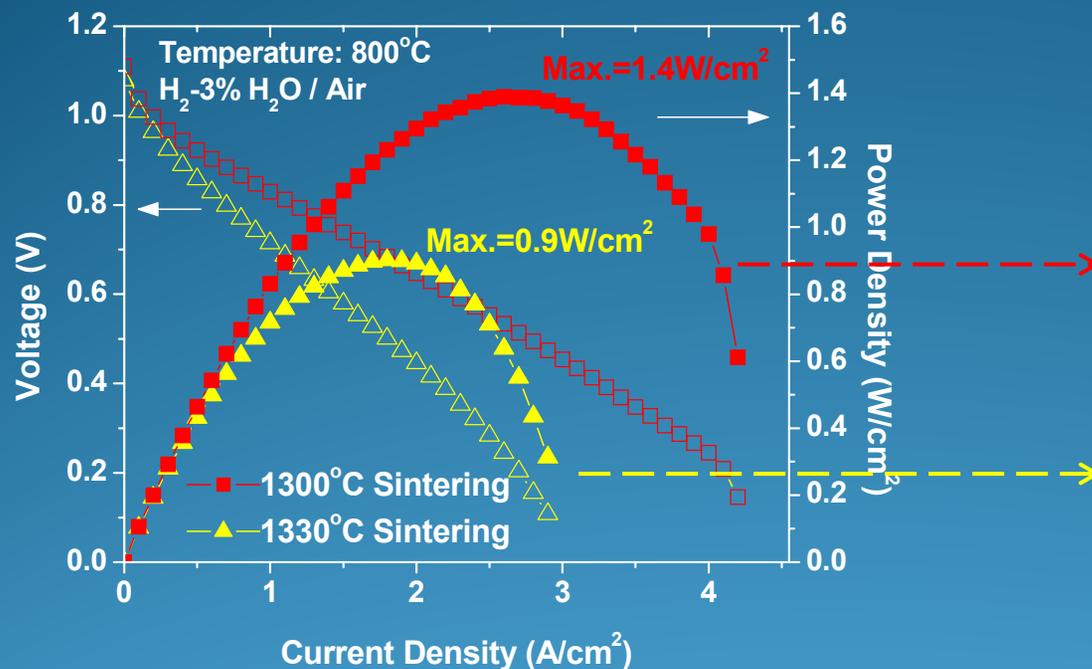
## ❖ Cathode Current Collection : Ag Mesh vs. Pt Mesh



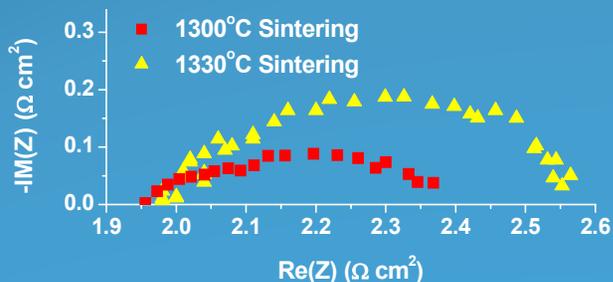
- Ag is softer than Pt at operating temperature.  
⇒ Higher Interfacial Contact Area ⇒ Lower **Ohmic Electrode Resistance**
- In SOFC stacks, the effect of the contacts between the electrode and interconnects can be substantial.

# POLARIZATION ANALYSIS #3: CATHODE MICROSTRUCTURE

## ❖ Sintering Temperature: 1300°C vs. 1330°C



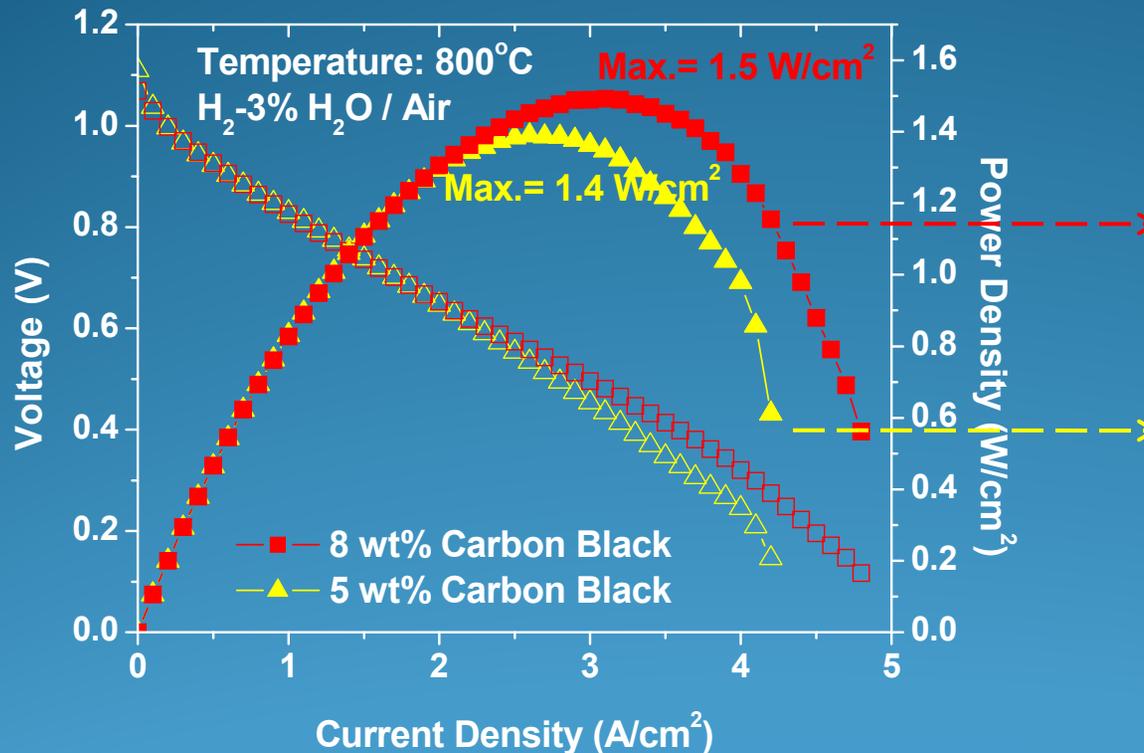
### Impedance Spectroscopy



- No Significant Change in Anode Microstructure.
- Less Sintering of Cathode ⇒ Low **Activation Polarization**

# POLARIZATION ANALYSIS #4: ANODE POROSITY

## ❖ Pore Former in Anode: 8 wt% C vs. 5 wt% C



8 wt% Carbon Black  
 Anode Porosity: 37%

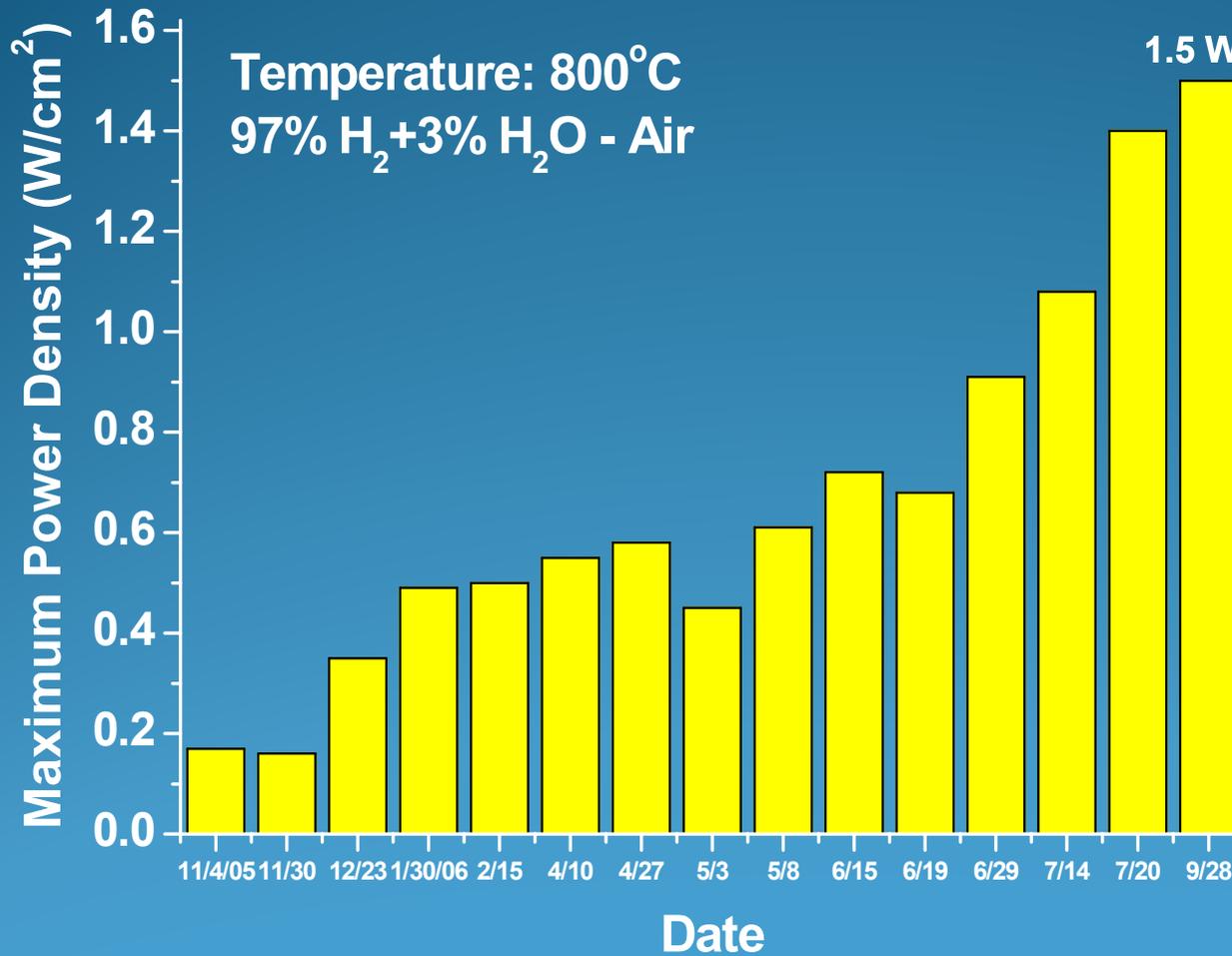
$i_{as} = 5.77 \text{ A/cm}^2$   
 $D_{H_2-H_2O}^{eff} = 0.23 \text{ cm}^2\text{s}^{-1}$

5 wt% Carbon Black  
 Anode Porosity: 32%

$i_{as} = 4.97 \text{ A/cm}^2$   
 $D_{H_2-H_2O}^{eff} = 0.19 \text{ cm}^2\text{s}^{-1}$

■ Increased Anode Porosity  
 ⇒ Low Anode Concentration Polarization

# CELL PERFORMANCE IMPROVEMENT CHART



Comparable to the State-of-the-Art Cells Fabricated with Multiple Firing Steps

# FUEL UTILIZATION TEST : BACKGROUND

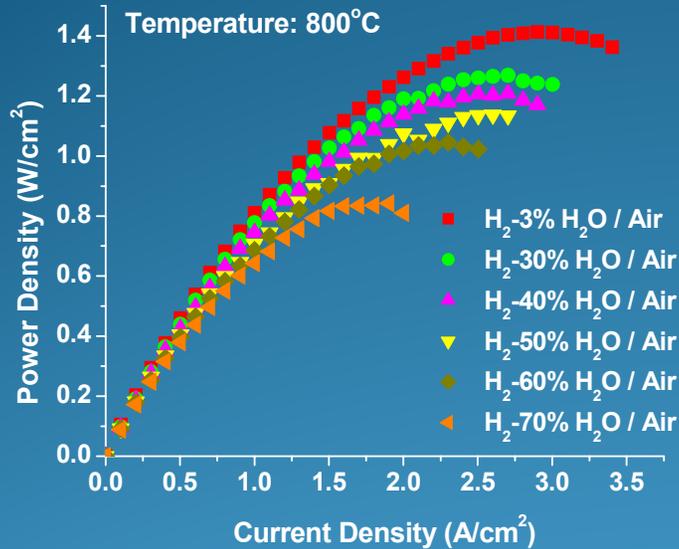
## ■ Fuel Utilization

$$U_t = \frac{\text{Molar flow rate of reactants consumed in a cell}}{\text{Molar flow rate of reactants supplied into the cell}} = \frac{\dot{N}_{consumed}}{\dot{N}_{in}}$$

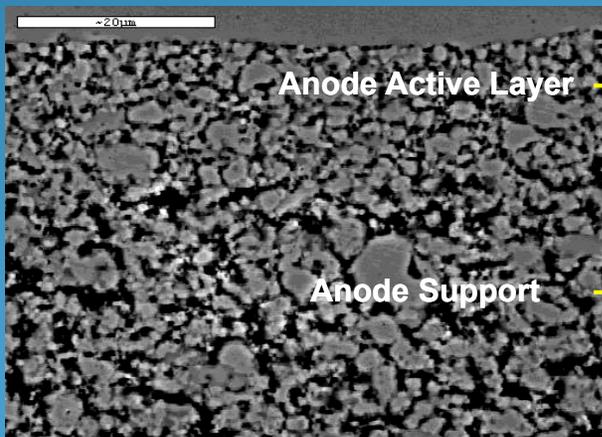
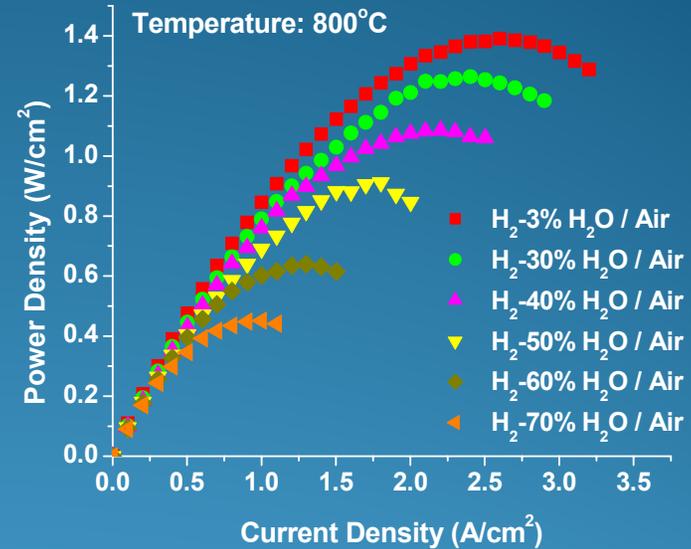
- Fuel utilization increases along the flow path over the electrode surface.  
⇐ Fuels are consumed and products are formed along the flow path.
- Cell Performance Loss near Exit (High Fuel Utilization)
  - Loss of Nernst Potential
  - Anodic Activation Polarization
  - Anodic Concentration Polarization
- Simulate the effect of practical fuel utilization on single cell performance by increasing H<sub>2</sub>O content in fuel

# FUEL UTILIZATION TEST : EFFECT OF ANODE ACTIVE LAYER

## With Anode Active Layer



## Without Anode Active Layer



Anode Active Layer

Porosity = 26%  
Avg. Grain Size = 1.3 µm  
Avg. Pore Size = 0.7 µm

Anode Support

Porosity = 37%  
Avg. Grain Size = 4.3 µm  
Avg. Pore Size = 2.6 µm

Fuel Compositions	Max. Power Density (W/cm <sup>2</sup> )	
	With AAL	Without AAL
H <sub>2</sub> - 3% H <sub>2</sub> O	1.41	1.40
H <sub>2</sub> - 30% H <sub>2</sub> O	1.27	1.25
H <sub>2</sub> - 50% H <sub>2</sub> O	1.17	0.91
H <sub>2</sub> - 70% H <sub>2</sub> O	0.84	0.45

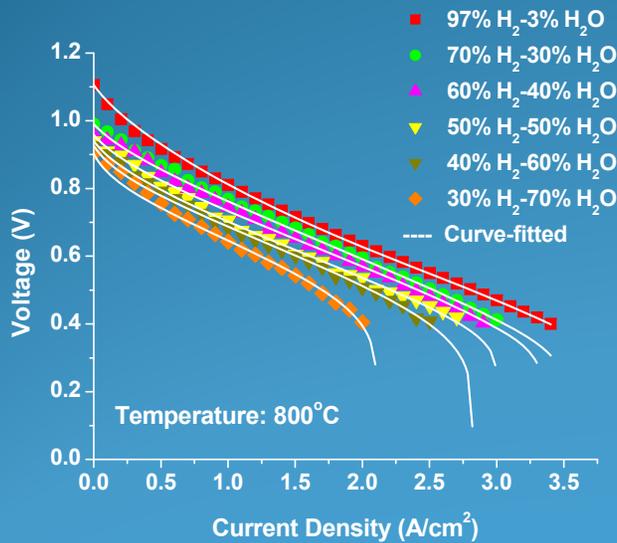
# FUEL UTILIZATION TEST : POLARIZATION MODELING

$$E_c = E_o - i R_i - \frac{2RT}{F} \ln \left\{ \frac{1}{2} \left[ \left( \frac{i}{i_o} \right) + \sqrt{\left( \frac{i}{i_o} \right)^2 + 4} \right] \right\} + \frac{RT}{4F} \ln \left( 1 - \frac{i}{i_{cs}} \right) + \frac{RT}{2F} \ln \left( 1 - \frac{i}{i_{as}} \right) - \frac{RT}{2F} \ln \left( 1 + \frac{p_{H_2}^o i}{p_{H_2O}^o i_{as}} \right)$$

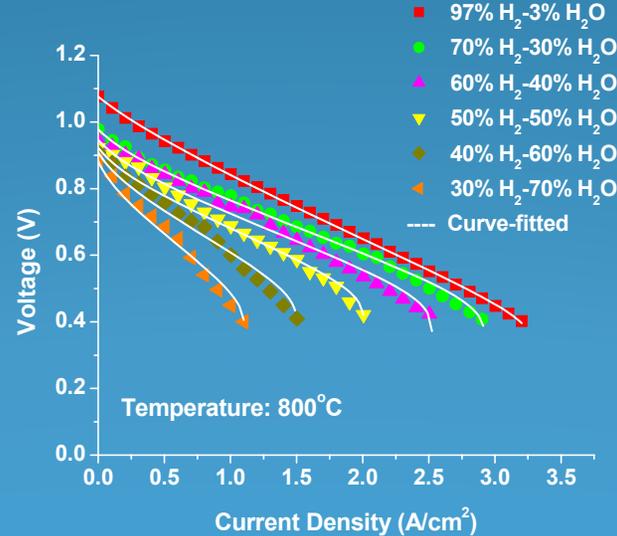
$R_i, D^{eff}_{O_2-N_2}$  : Independent of Fuel Composition

$D^{eff}_{H_2-H_2O}$  : Independent of  $H_2/H_2O$  ratio (Kinetic Theory of Gases)\*

## With Anode Active Layer



## Without Anode Active Layer



Fuel Compositions	Exchange Current Density (A/cm <sup>2</sup> )	
	With AAL	Without AAL
H <sub>2</sub> - 3% H <sub>2</sub> O	0.87	0.98
H <sub>2</sub> - 30% H <sub>2</sub> O	0.84	0.78
H <sub>2</sub> - 40% H <sub>2</sub> O	0.82	0.70
H <sub>2</sub> - 50% H <sub>2</sub> O	0.79	0.57
H <sub>2</sub> - 60% H <sub>2</sub> O	0.75	0.43
H <sub>2</sub> - 70% H <sub>2</sub> O	0.53	0.22

\* R. Byron Bird, Warren E. Stewart, Edwin N. Lightfoot, Transport Phenomena, John Wiley & Sons (1960)

# FUEL UTILIZATION TEST : PERFORMANCE ANALYSIS

## ❖ Activation Polarization:

### ▪ At **Low Fuel Utilization** (H<sub>2</sub>-3% H<sub>2</sub>O)

⇒ Dominated by **Cathode**

(No Difference in **Cell Performance** and **Exchange Current Density** due to Anode Active Layer at Low Fuel Utilization)

$$\eta_{act} \approx \eta_{act,c} = \frac{2RT}{F} \ln \left\{ \frac{1}{2} \left[ \left( \frac{i}{i_{0,c}} \right) + \sqrt{\left( \frac{i}{i_{0,c}} \right)^2 + 4} \right] \right\}$$

$$i_{0,a} \gg i_{0,c} \quad i_o \approx i_{0,c}$$

### ▪ At **High Fuel Utilization**

⇒ Anodic Activation Polarization Increases

⇒ Cathodic Activation Polarization: Independent of Fuel Composition

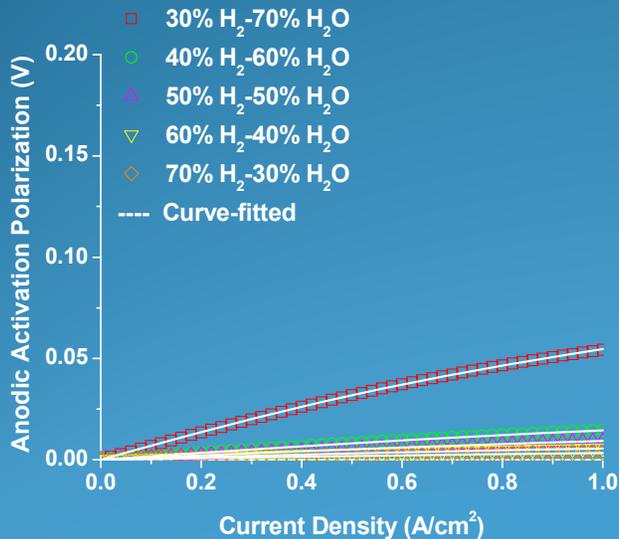
$$\eta_{act,a} \approx \eta_{act} - \eta_{act,c} = \frac{2RT}{F} \ln \left\{ \frac{1}{2} \left[ \left( \frac{i}{i_{0,a}} \right) + \sqrt{\left( \frac{i}{i_{0,a}} \right)^2 + 4} \right] \right\}$$

# FUEL UTILIZATION TEST : PERFORMANCE ANALYSIS

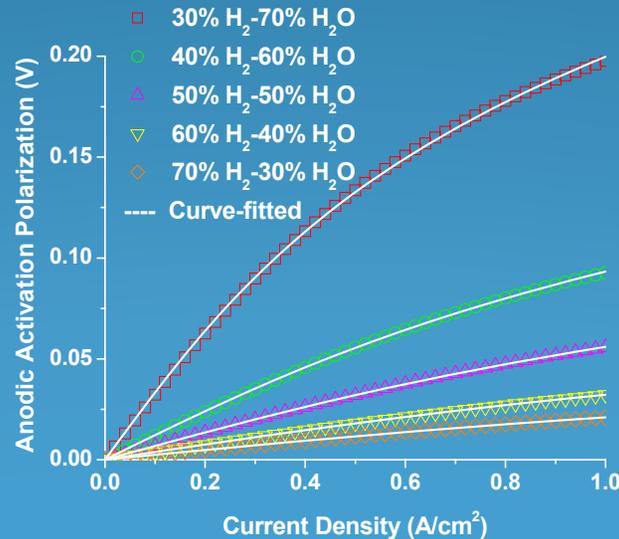
## ❖ Anodic Activation Polarization:

$$\eta_{act,a} = \frac{2RT}{F} \ln \left\{ \frac{1}{2} \left[ \left( \frac{i}{i_{0,a}} \right) + \sqrt{\left( \frac{i}{i_{0,a}} \right)^2 + 4} \right] \right\}$$

### With Anode Active Layer



### Without Anode Active Layer



Fuel Compositions	Anodic Exchange Current Density (A/cm <sup>2</sup> )	
	With AAL	Without AAL
H <sub>2</sub> – 30% H <sub>2</sub> O	26.73	4.18
H <sub>2</sub> – 40% H <sub>2</sub> O	15.68	2.70
H <sub>2</sub> – 50% H <sub>2</sub> O	9.47	1.53
H <sub>2</sub> – 60% H <sub>2</sub> O	6.02	0.88
H <sub>2</sub> – 70% H <sub>2</sub> O	1.55	0.34
<b>Cathodic Exchange Current Density (A/cm<sup>2</sup>)</b>	0.87	0.98

# FUEL UTILIZATION TEST : PERFORMANCE ANALYSIS

## ❖ Anodic Concentration Polarization:

$$\eta_{conc,a} = \frac{RT}{4F} \ln\left(\frac{p_{O_2}^{i,a}}{p_{O_2}^{o,a}}\right)$$

$p_{O_2}^{i,a}$ : Oxygen partial pressure at anode-electrolyte interface

$p_{O_2}^{o,a}$ : Oxygen partial pressure outside anode surface

Determined by Local  $H_2$ - $H_2O$  Equilibrium:  $p_{O_2} = \frac{p_{H_2O}^2}{Kp_{H_2}^2}$

### Calculation of $p_{O_2}^{i,a}$

In Steady State,  $J_{H_2} = J_{H_2O} = \frac{i}{2F}$        $J_{H_2} + J_{H_2O} = 0$

$$J_{H_2} = -D_{H_2-H_2O}^{eff} \nabla n_{H_2} = -D_{H_2-H_2O}^{eff} \frac{p_{H_2}^i - p_{H_2}^o}{RTl_a}$$



$$p_{H_2}^i = p_{H_2}^o - \frac{RTl_a}{2FD_{H_2-H_2O}^{eff}} i$$

$$J_{H_2O} = -D_{H_2-H_2O}^{eff} \nabla n_{H_2O} = -D_{H_2-H_2O}^{eff} \frac{p_{H_2O}^o - p_{H_2O}^i}{RTl_a}$$



$$p_{H_2O}^i = p_{H_2O}^o + \frac{RTl_a}{2FD_{H_2-H_2O}^{eff}} i$$

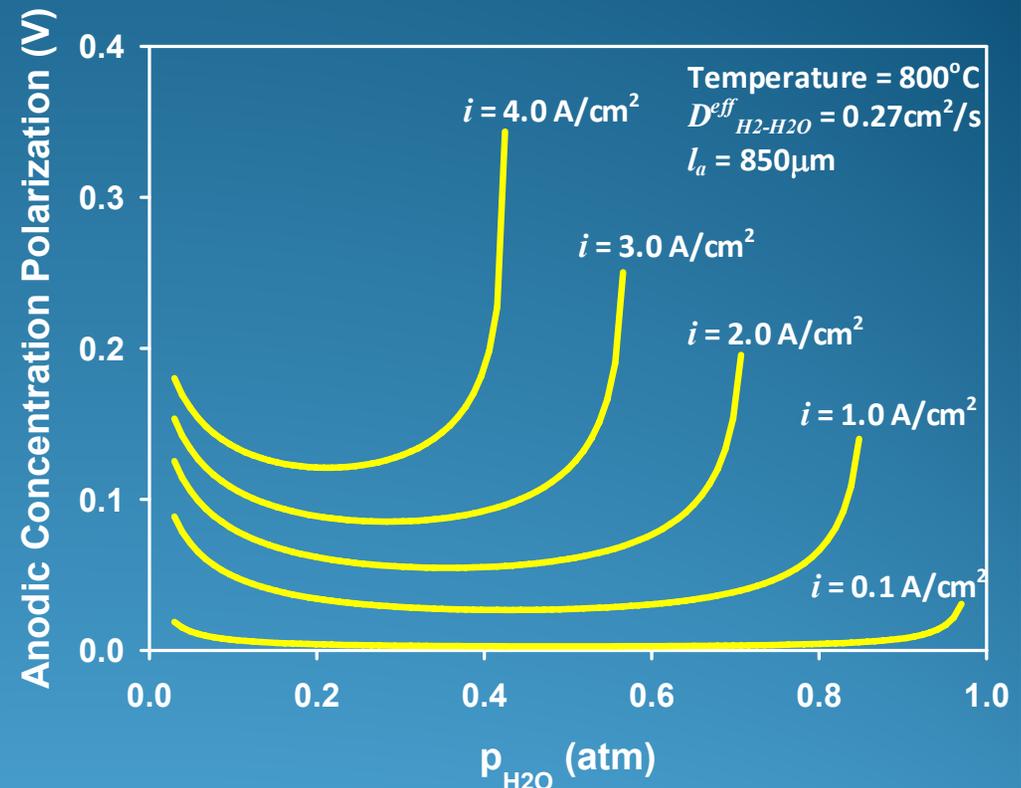


# FUEL UTILIZATION TEST : PERFORMANCE ANALYSIS

## ❖ Anodic Concentration Polarization:

$$\eta_{conc,a} = \frac{RT}{4F} \ln \left( \frac{p_{O_2}^{i,a}}{p_{O_2}^{o,a}} \right)$$

$$= \frac{RT}{2F} \ln \left[ \frac{\left( p_{H_2O}^{o,a} + \frac{RTl_a}{2FD_{H_2-H_2O}^{eff}} i \right) p_{H_2}^{o,a}}{\left( p_{H_2}^{o,a} - \frac{RTl_a}{2FD_{H_2-H_2O}^{eff}} i \right) p_{H_2O}^{o,a}} \right]$$

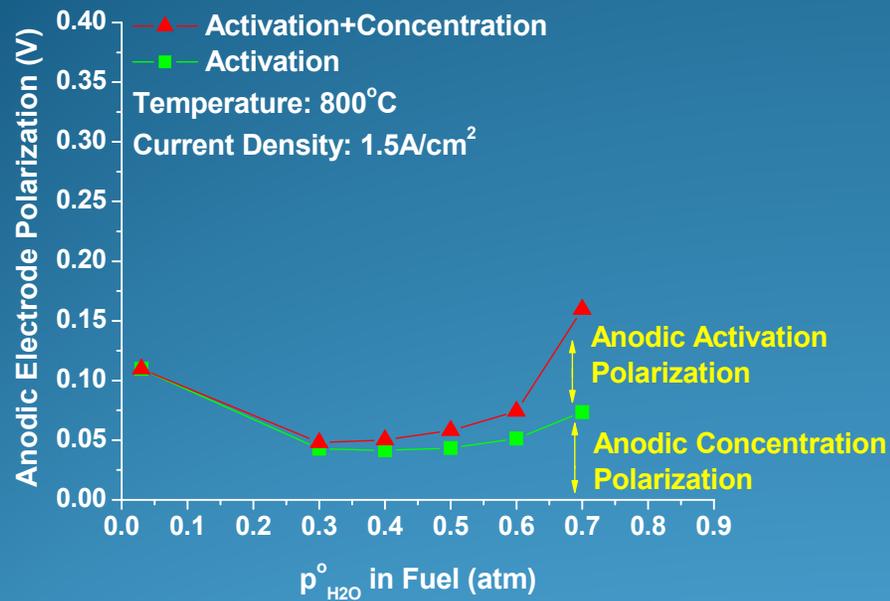


- Anodic concentration polarization is low when the fuel is **in the intermediate H<sub>2</sub>O partial pressure region**.
- Anode active layer had no significant effect on anodic concentration polarization.

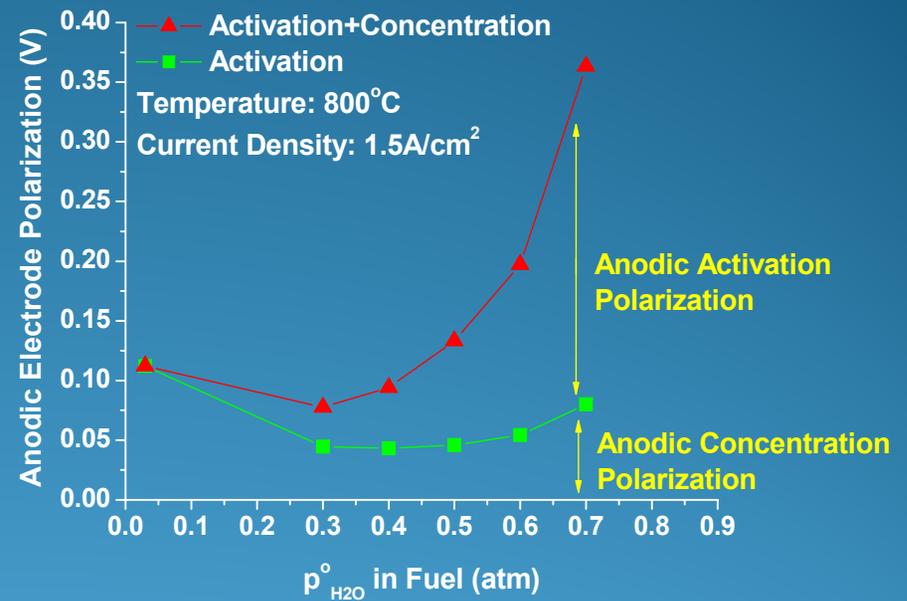
# FUEL UTILIZATION TEST : POLARIZATION ANALYSIS

## ❖ Anodic Electrode Polarization Loss:

### With Anode Active Layer



### Without Anode Active Layer



- **Activation Polarization: Dominant Loss at High Fuel Utilization**  
⇒ Significantly Reduced by **Anode Active Layer**

# ADVANCED CATHODE INVESTIGATION : BACKGROUND

- Cathode Polarization Loss : Major Difficulty in Lowering the Operating Temperature
- A site-doped Lanthanum Cobaltite
  - High Catalytic Activity and Mixed Electronic-Ionic Conductivity
  - High Thermal Expansion Coefficient
  - Solid State Reaction with YSZ at Low Temperature
- A site-doped Lanthanum Ferrite
  - High Catalytic Activity and Mixed Electronic-Ionic Conductivity
  - Adjustable Thermal Expansion
  - No Solid State Reaction with YSZ up to 1400°C
  - Diffusion of  $Zr^{4+}$  into Lanthanum Ferrite : Doped Ceria Interlayer
- Calcium-doped Lanthanum Ferrite
  - Defect Model
  - Thermogravimetry Measurements
  - Electrical Conductivity Measurements  $\Rightarrow$  Hole Mobility

$\left. \begin{array}{l} \text{Defect Model} \\ \text{Thermogravimetry Measurements} \end{array} \right\} p_{O_2} \text{- Weight Relationship} \\ \Rightarrow \text{Equilibrium Defect Concentration}$

# ADVANCED CATHODE INVESTIGATION : POINT DEFECT MODEL

## ❖ Point Defect Model for $(La_{0.8}Ca_{0.2})_{0.95}FeO_{3-\delta}$

A-site	B-site	O-site
$La_{La}^x$	$Fe_{Fe}^x$	$O_O^x$
$Ca'_{La} = 0.2 \times 0.95$	$Fe'_{Fe}$	$V_O^{**}$
$V_{La}''' = 0.05$	<del><math>Fe'_{Fe} \approx 0</math></del>	
	<del><math>V_{Fe}''' \approx 0</math></del>	

### ▪ Oxygen Incorporation Reaction



### ▪ Charge Disproportionation Reaction



### ▪ Schottky Equilibrium Reaction



# ADVANCED CATHODE INVESTIGATION : POINT DEFECT MODEL

- **Charge Neutrality Condition**

$$2[V_O^{\bullet\bullet}] + [Fe_{Fe}^{\bullet}] = [Ca'_{La}] + 3[V_{La}^{\bullet\bullet\bullet}] \implies [Fe_{Fe}^{\bullet}] = [Ca'_{La}] + 3[V_{La}^{\bullet\bullet\bullet}] - 2[V_O^{\bullet\bullet}]$$

$= 0.2 \quad 0.95 \quad = 0.05$

- **A-site Restriction**

$$[La^x_{La}] + [Ca'_{La}] + [V_{La}^{\bullet\bullet\bullet}] = 1$$

- **B-site Restriction**

$$[Fe^x_{Fe}] + [Fe_{Fe}^{\bullet}] = 1 \implies [Fe^x_{Fe}] = 1 - [Fe_{Fe}^{\bullet}] = 1 - [Ca'_{La}] - 3[V_{La}^{\bullet\bullet\bullet}] + 2[V_O^{\bullet\bullet}]$$

$= 0.2 \times 0.95 \quad = 0.05$

- **O-site Restriction**

$$[O^x_O] + [V_O^{\bullet\bullet}] = 3 \implies [O^x_O] = 3 - [V_O^{\bullet\bullet}]$$

- **Mass Action Coefficient for Oxygen Exchange Reaction**

$$K_{ox} = \frac{[O^x_O][Fe_{Fe}^{\bullet}]^2}{[V_O^{\bullet\bullet}][Fe^x_{Fe}]^2 p_{O_2}^{1/2}}$$

$$K_{ox} = \frac{(3 - [V_O^{\bullet\bullet}])([Ca'_{La}] + 3[V_{La}^{\bullet\bullet\bullet}] - 2[V_O^{\bullet\bullet}])^2}{[V_O^{\bullet\bullet}](1 - [Ca'_{La}] - 3[V_{La}^{\bullet\bullet\bullet}] + 2[V_O^{\bullet\bullet}])^2} \cdot \frac{1}{p_{O_2}^{1/2}}$$

# ADVANCED CATHODE INVESTIGATION : POINT DEFECT MODEL

- Relationship between  $p_{O_2}$  and Weight of  $(La_{0.8}Ca_{0.2})_{0.95}FeO_{3-\delta}$

$$K_{ox} = \frac{(3 - [V_O^{**}])([Ca_{La}'] + 3[V_{La}'''] - 2[V_O^{**}])^2}{[V_O^{**}](1 - [Ca_{La}'] - 3[V_{La}'''] + 2[V_O^{**}])^2} \cdot \frac{1}{p_{O_2}^{1/2}}$$

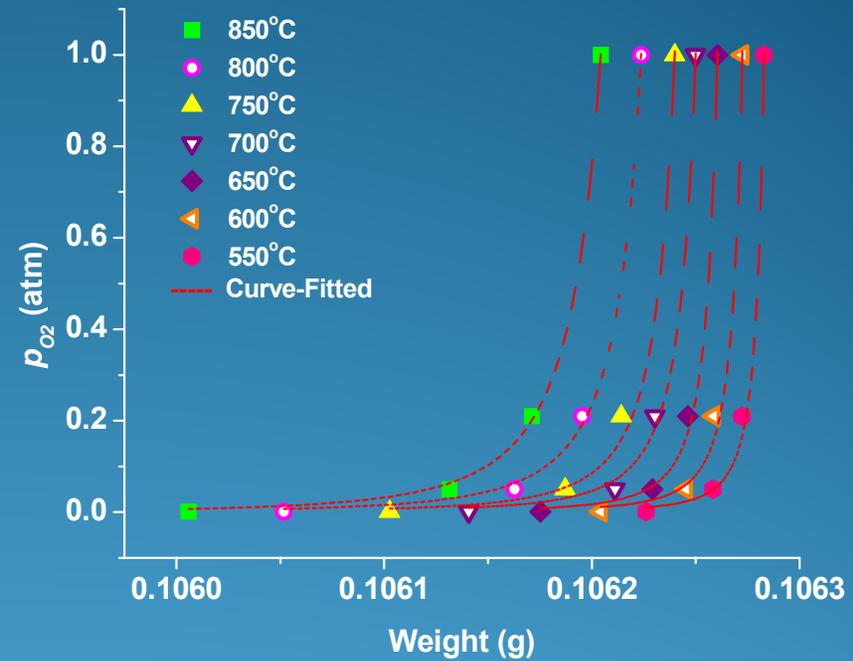
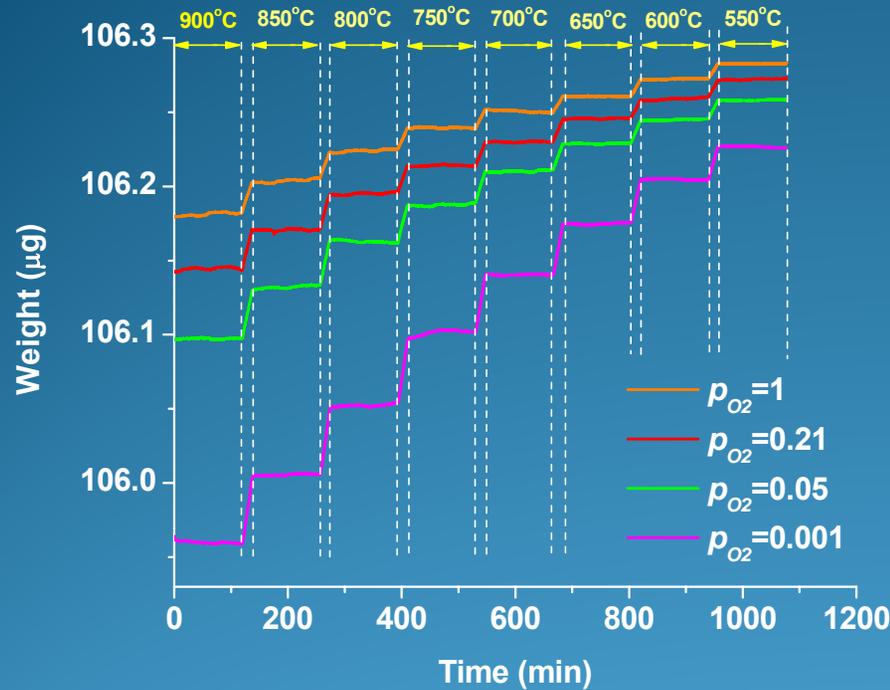
$$M_{LCF} = n \times m_{LCF} = n \times [(0.8 \times m_{La} + 0.2 \times m_{Ca}) \times 0.95 + m_{Fe} + (3 - \delta) \times m_O]$$

$$\delta = [V_O^{**}] = 13.56 - 0.0625 \times \frac{M_{LCF}}{n}$$

$$p_{O_2} = \frac{1}{K_{ox}^2} \times \frac{\left(-10.56 + 0.0625 \times \frac{M_{LCF}}{n}\right)^2 \left(-26.78 + 0.125 \times \frac{M_{LCF}}{n}\right)^4}{\left(13.56 - 0.0625 \times \frac{M_{LCF}}{n}\right)^2 \left(27.78 + 0.125 \times \frac{M_{LCF}}{n}\right)^4}$$

# ADVANCED CATHODE INVESTIGATION : THERMOGRAVIMETRY

## Thermogravimetry Measurements



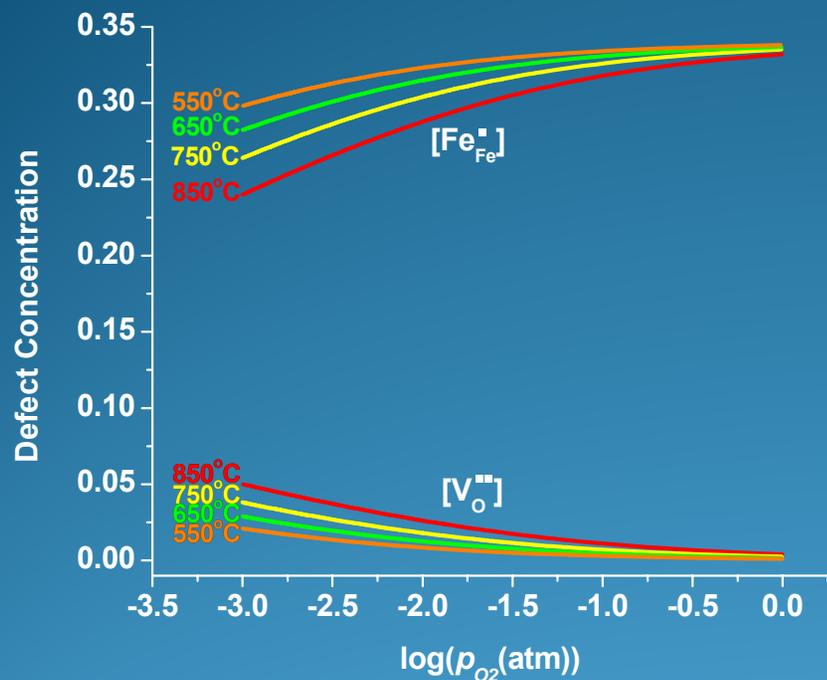
$$p_{O_2} = \frac{1}{K_{ox}^2} \times \frac{\left(-10.56 + 0.0625 \times \frac{M_{LCF}}{n}\right)^2 \left(-26.78 + 0.125 \times \frac{M_{LCF}}{n}\right)^4}{\left(13.56 - 0.0625 \times \frac{M_{LCF}}{n}\right)^2 \left(27.78 + 0.125 \times \frac{M_{LCF}}{n}\right)^4}$$

$n = 0.00049 \text{ mol}$

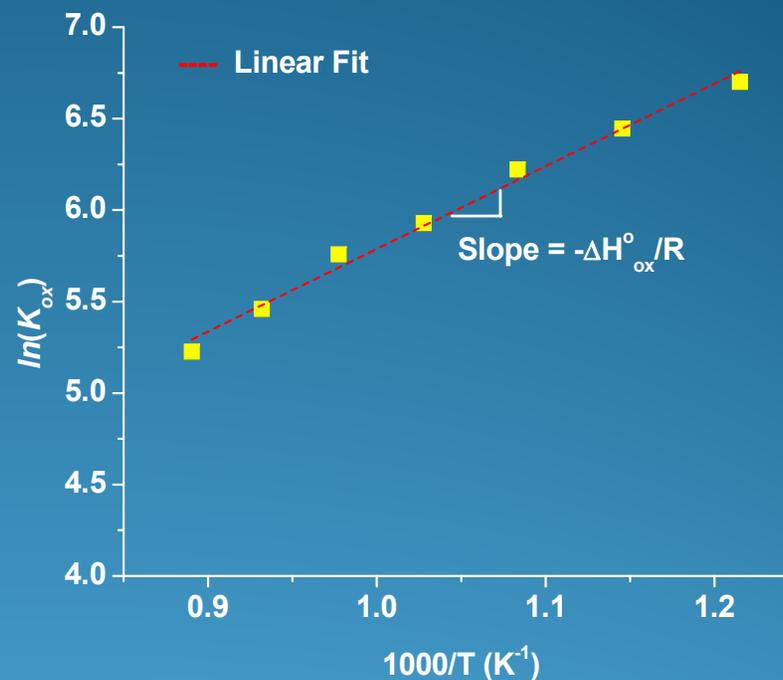
Temperature	850°C	800°C	750°C	700°C	650°C	600°C	550°C
$K_{ox}$	186	235	317	376	504	630	813

# ADVANCED CATHODE INVESTIGATION : DEFECT EQUILIBRIUM

## Equilibrium Defect Concentration



## $\ln(K_{\text{ox}})$ vs. $1/T$



### ➤ High $p_{\text{O}_2}$ range

- $[\text{V}_{\text{O}}^{\bullet\bullet}]$ : low
- Charge neutrality is maintained by hole formation.

### ➤ Low $p_{\text{O}_2}$ range

- $[\text{V}_{\text{O}}^{\bullet\bullet}]$ : high
- Hole concentration decreases.

$$\ln(K_{\text{ox}}) = -\frac{\Delta H_{\text{ox}}^{\circ}}{RT} + \frac{\Delta S_{\text{ox}}^{\circ}}{R}$$

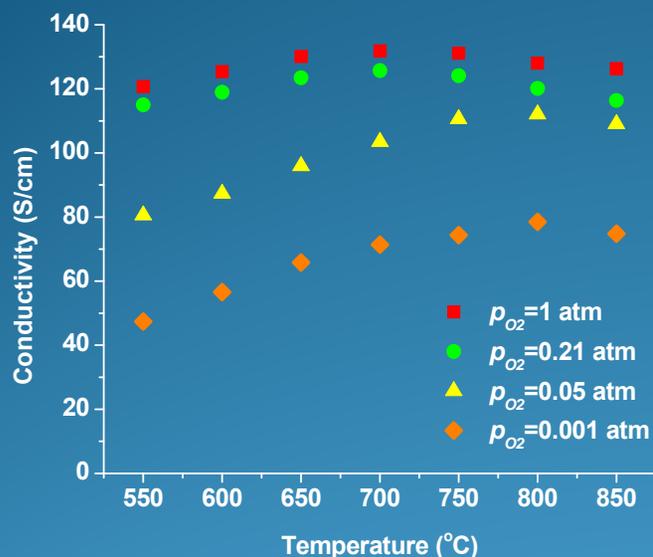
$$\Delta H_{\text{ox}}^{\circ} = -37.5 \text{ kJ mol}^{-1}$$

$$\Delta S_{\text{ox}}^{\circ} = 10.6 \text{ J mol}^{-1} \text{ K}^{-1}$$

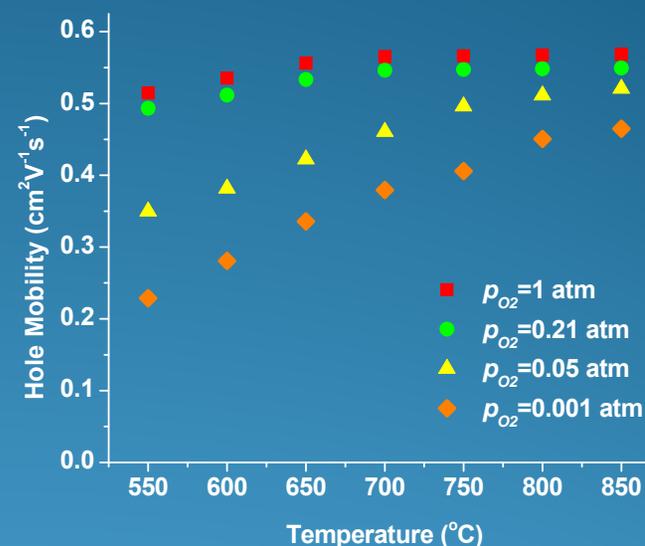
# ADVANCED CATHODE INVESTIGATION : ELECTRICAL CONDUCTIVITY

## Electrical Conductivity Measurements

### Conductivity vs. Temperature



### Mobility vs. Temperature



$$\sigma_t = \sigma_e + \sigma_i \approx \sigma_e$$

$$\sigma_p = \frac{[Fe_{Fe}^{\bullet}]}{V} q \mu_p$$

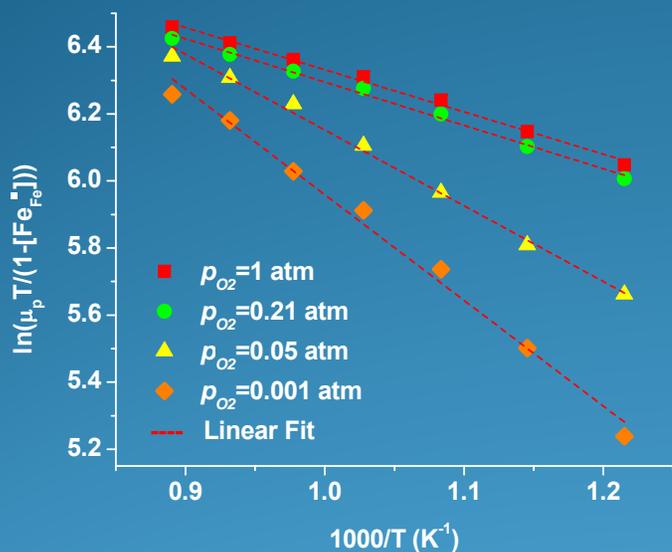
- $p_{O_2} \uparrow \Rightarrow \sigma_e \uparrow$  : **p-type conductor**
- Low Temperature: **Thermally Activated Behavior**  
(Small Polaron Hopping)
- High Temperature: **Decrease in Hole Concentration**

- $p_{O_2} \downarrow, T \uparrow \Rightarrow [V_{O}^{\bullet}] \uparrow$
- Hopping Conduction Via **Fe<sup>4+</sup>-O-Fe<sup>3+</sup>** Chain
- V<sub>O</sub><sup>•</sup>** : **Scattering Centers or Random Traps** for Electrons

# ADVANCED CATHODE INVESTIGATION : CONDUCTION MECHANISM

## Adiabatic Small Polaron Hopping

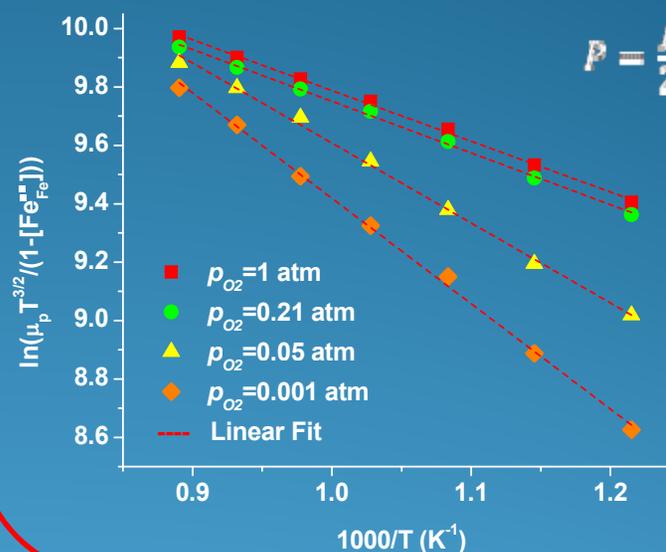
$$\mu_p = \frac{(1 - [\text{Fe}_{\text{Fe}}^{\bullet}])qa^2v}{kT} \exp\left(-\frac{E_A}{kT}\right)$$



## Non-adiabatic Small Polaron Hopping

$$\mu_p = \frac{(1 - [\text{Fe}_{\text{Fe}}^{\bullet}])qa^2}{kT} P \exp\left(-\frac{E_A}{kT}\right)$$

$$P = \frac{J^2}{2\hbar} \left(\frac{\pi}{E_A kT}\right)^{1/2}$$



p <sub>O2</sub> (atm)	Adiabatic Case		Non-adiabatic Case	
	Activation Energy (eV)	R <sup>2</sup>	Activation Energy (eV)	R <sup>2</sup>
1	0.108	0.99353	0.149	0.99788
0.21	0.111	0.99344	0.152	0.99776
0.05	0.195	0.99423	0.235	0.99673
0.001	0.272	0.98927	0.310	0.99801

## SUMMARY

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- Successfully developed single-step un-constrained co-firing of the solid oxide fuel cell @ 1300°C.
- Modeled cell performance.
- Achieved maximum power density of 1.50 W/cm<sup>2</sup> at 800°C and 0.87 W/cm<sup>2</sup> at 700°C with humidified hydrogen (3% H<sub>2</sub>O) and air.
- Simulated the effect of practical fuel utilization on single cell performance.
- Improved cell performance at high fuel utilization by employing anode active layer.
- Investigated defect chemistry and electrical conduction mechanism of novel cathode material (calcium-doped lanthanum ferrite).

## FUTURE WORK

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- Employ advanced cathode material in co-firing process.
- Analyze performance at low operating temperature (600-700°C).

## ■ Journal Papers

1. Kyung Joong Yoon, Wenhua Huang, Guosheng Ye, Srikanth Gopalan, Uday B. Pal, Donald A. Seccombe, Jr., "Electrochemical Performance of Solid Oxide Fuel Cells (SOFCs) Manufactured by Single Step Co-firing Process," *Journal of the Electrochemical Society*, 154 (4) B389 (2007).
2. Kyung Joong Yoon, Srikanth Gopalan, Uday B. Pal, "Effect of Fuel Composition on Performance of Single Step Co-fired Solid Oxide Fuel Cells (SOFCs)," *Journal of the Electrochemical Society*, 154 (10) B1080 (2007).
3. Kyung Joong Yoon, Peter Zink, Srikanth Gopalan, Uday B. Pal, "Polarization Measurements on Single Step Co-fired Solid Oxide Fuel Cells (SOFCs)," *Journal of Power Sources*, 172 (1) 39 (2007).
4. Kyung Joong Yoon, Srikanth Gopalan, Uday B. Pal, "Effect of Anode Active Layer on Performance of Single Step Co-fired Solid Oxide Fuel Cells (SOFCs) at High Fuel Utilizations," *Journal of the Electrochemical Society*, 155(6) B610 (2008).
5. Kyung Joong Yoon, Srikanth Gopalan, Uday B. Pal, "Analysis of Electrochemical Performance of Solid Oxide Fuel Cells (SOFCs) Using Polarization Modeling and Impedance Measurements," *Journal of the Electrochemical Society*, 156 (3) B311 (2008).
6. Kyung Joong Yoon, Guosheng Ye, Srikanth Gopalan, Uday B. Pal, "Cost-effective Single Step Co-firing Technique for Manufacturing Solid Oxide Fuel Cells (SOFCs) using High Shear Compaction (HSC) Anode," *Journal of Fuel Cell Science and Technology*, Accepted (2008).
7. Soobhankar Pati, Kyung Joong Yoon, Uday B. Pal, "Solid Oxide Electrolyte Electrolyzer with Liquid Metal Anode for Production of Hydrogen and Syn-Gas from Waste and Steam," submitted (2009).
8. Kyung Joong Yoon, Peter Zink, Larry Pederson, Srikanth Gopalan, Uday B. Pal, "Defect Chemistry and Electrical Properties of  $(La_{0.8}Ca_{0.2})_{0.95}FeO_{3.8}$ ," in preparation (2009).

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1. Kyung Joong Yoon, Peter Zink, Srikanth Gopalan, Uday B. Pal, "Polarization Analysis in Single Step Co-fired Solid Oxide Fuel Cells (SOFCs)," *Materials Research Society Symposium Proceedings of the Fall 2006 Meeting*, Vol. 972, AA 10-02 (2007).
2. Peter A. Zink, Kyung Joong Yoon, Wenhua Huang, Srikanth Gopalan, Uday B. Pal, Donald A. Seccombe, Jr., "Refractory Cathode Investigation for Single Step Co-fired Solid Oxide Fuel Cells (SOFCs)," *Materials Research Society Symposium Proceedings of the Fall 2006 Meeting*, Vol. 972, AA 03-12 (2007).
3. Kyung Joong Yoon, Peter Zink, Uday B. Pal, Srikanth Gopalan, "High Performance Low Cost Co-fired Solid Oxide Fuel Cells (SOFCs)," *ECS Transactions*, Vol. 7 (1) 579 (2007).
4. Kyung Joong Yoon, Srikanth Gopalan, Uday B. Pal, "Anode Polarization Effects in Single Step Co-fired Solid Oxide Fuel Cells (SOFCs)," *ECS Transactions*, Vol. 7 (1) 565 (2007).
5. Peter Zink, Kyung Joong Yoon, Wenhua Huang, Uday B. Pal, Srikanth Gopalan, "Refractory Cathode Investigation for Single Step Co-fired Solid Oxide Fuel Cells (SOFCs)," *ECS Transactions*, Vol. 7 (1) 399 (2007).
6. Kyung Joong Yoon, Srikanth Gopalan, Uday B. Pal, "Effect of Anode Active Layer on Performance of Single Step Co-fired Solid Oxide Fuel Cells (SOFCs)," *ECS Transactions*, Vol. 13 (26) 249 (2008).
7. Kyung Joong Yoon, Srikanth Gopalan, Uday B. Pal, "Electrochemical Performance of Single Step Co-Fired Solid Oxide Fuel Cells (SOFCs) Analyzed Using Polarization Modeling and Impedance Spectroscopy," *Materials Research Society Symposium Proceedings of the Fall 2008 Meeting*, Vol. 1126, S10-02 (2008).
8. Peter A. Zink, Kyung Joong Yoon, Uday B. Pal, Srikanth Gopalan, "Electrical Performance of Calcium-doped Lanthanum Ferrite for Use in Single Step Co-Fired Solid Oxide Fuel Cells (SOFCs)," *Materials Research Society Symposium Proceedings of the Fall 2008 Meeting*, Vol. 1126, S11-02 (2008).

Thank you!